

GROWTH OF CRACK-FREE THICK AlGa_N LAYER AND ITS APPLICATION TO GaN-BASED LASER DIODE

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ABSTRACT

In the field of group-III nitrides, hetero-epitaxial growth has been one of the most important key technologies. A thick layer of AlGa_N alloy with higher AlN molar fraction is difficult to grow on sapphire substrate, because the alloy layer is easily cracked. It is thought that one cause of generating cracks is a large lattice mismatch between an AlGa_N and a GaN, when AlGa_N is grown on the underlying GaN layer. We have achieved crack-free Al_{0.07}Ga_{0.93}N layer with the thickness of more than 1 μm using underlying Al_{0.05}Ga_{0.95}N layer. The underlying Al_{0.05}Ga_{0.95}N layer was grown directly on sapphire by using the low-temperature-deposited buffer layer (LT-buffer layer). Since a lattice mismatch between the underlying Al_{0.05}Ga_{0.95}N layer and upper Al_{0.07}Ga_{0.93}N layer is relatively small, the generation of cracks is thought to be suppressed. This technology is applied to a GaN-based laser diode structure, in which thick n-Al_{0.07}Ga_{0.93}N cladding layer grown on the Al_{0.05}Ga_{0.95}N layer, improves optical confinement and single-lobe far field pattern in vertical direction.

INTRODUCTION

GaN-based semiconductor lasers are promising for light sources of high-density optical data storage systems, because of their short wavelength of around 400nm. However, we have worked with much effort for a long time in order to obtain a high quality crystal, because of the lack of substrates lattice-matched to this material system. In 1986, on the growth method using a LT-buffer layer[1] on sapphire substrate, which has a large lattice-mismatch of about 16%, GaN film with an excellent flatness and crystallinity was successfully obtained. Further, p-type low resistance crystal was realized by Mg-doping and electron-beam irradiation [2], or anneal process [3]. After such progress, high brightness blue and green light emitting diodes (LEDs) have been realized [4],[5]. More recently, several groups have achieved continuous-wave (CW) operation of violet-blue laser diodes(LD)[6]-[9]. It is thought that the GaN-based LD for the practical use will be commercially available in the near future.

However, the further improvements concerning with a crystalline quality and device performance are required against the practical use. In general, high quality GaN layer is grown on LT-buffer layer / sapphire substrate. When the AlGaN layers, commonly used as cladding layers in LDs, are grown on the GaN layer, the crack is often observed in the surface. This is thought to be due to the difference of lattice constants between GaN and AlGaN. Therefore, the thickness and AlN molar fraction of AlGaN layer, which are considerably important for the device performance, are greatly limited in order to avoid the crack generation.

The beam profiles of GaN-based diode laser in the direction perpendicular to the junction (vertical mode) are tend to be composed of multiple spots, which can be seen in the far field pattern [10],[11]. A poor optical confinement causes optical leakage from a waveguide layer to the underlying layer, such as an underlying n-GaN base layer. In order to suppress this optical leakage, the thickness of an AlGaN cladding layer should be much thicker than that usually embedded in a conventional structure. However it is quite difficult to grow crack-free thick AlGaN directly on GaN as mentioned above. Thus, the suppression of the crack in AlGaN is important subject for the improvement of device performance.

In this report, we demonstrate a crack-free $\text{Al}_{0.07}\text{Ga}_{0.93}\text{N}$ layer with the thickness of more than $1\mu\text{m}$ using underlying AlGaN layer grown directly on LT-buffer layer. We confirmed that the crystalline quality of AlGaN just on the LT-buffer layer is comparable as that of GaN. We also achieved a single-lobe far field pattern in the LD with the thick and crack-free n-AlGaN layer as a cladding

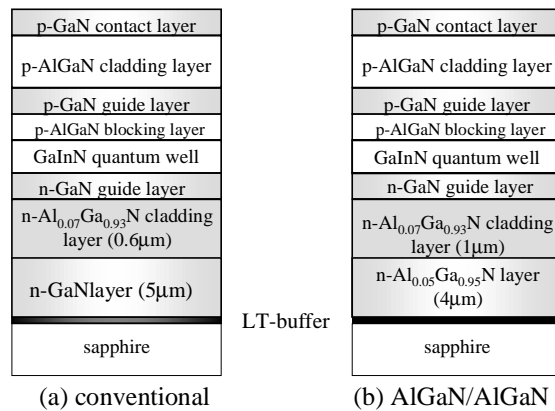


Figure 1: Schematic of laser structure with AlGaN/LT-buffer/sapphire.

layer.

CRYSTAL GROWTH

In a conventional laser structure, whose schematic diagram is shown in Fig. 1 (a), a n-GaN layer grown just on the LT-buffer layer is used as a base layer for laser structure, as well as a n-contact layer. However, this layer might cause the crack in n-AlGaN cladding layer grown on the layer, because of the tensile strain from the difference of lattice constants between GaN and AlGaN. We applied a new approach to overcome the problem that n-AlGaN is used as the base layer shown in Fig. 1 (b). We investigated a crystalline quality of n- $\text{Al}_{0.05}\text{Ga}_{0.95}\text{N}$ layer directly grown on the LT-buffer layer / sapphire substrate, using X-ray diffraction measurement and transmission electron microscopy (TEM). The FWHM of X-ray rocking curve was about 400 arcsec, and dislocation density from TEM image was $4 \times 10^9 \text{cm}^{-2}$, which values are comparable to those of GaN grown on LT-buffer layer. From these results, the n- $\text{Al}_{0.03}\text{Ga}_{0.97}\text{N}$ layer directly grown on the LT-buffer layer has a capability for the base layer of GaN-based laser diodes.

Laser structure was grown on c-face sapphire substrates by low-pressure metalorganic vapor phase epitaxy. In the new structure shown in Fig. 1 (b), instead of underlying n-GaN layer, a 4 μm thick n- $\text{Al}_{0.05}\text{Ga}_{0.95}\text{N}$ layer was directly grown on LT-buffer layer. After the growth of the underlying n- $\text{Al}_{0.05}\text{Ga}_{0.95}\text{N}$ layer, the other layers of laser structure were grown, which are 1 μm n-

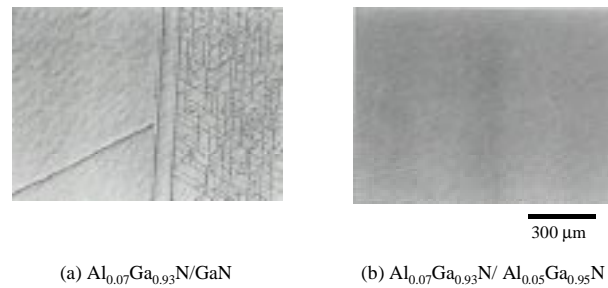


Figure 2: Surface morphologies of the laser structures

Al_{0.07}Ga_{0.93}N cladding layer, 0.1μm n-GaN guiding layer, 3 nm Ga_{0.9}In_{0.1}N / 10 nm Si-doped Ga_{0.98}In_{0.02}N 3QWs active layer with 15nm Al_{0.15}Ga_{0.85}N electron blocking layer, 0.1μm p-GaN guiding layer, 0.5μm p-Al_{0.07}Ga_{0.93}N layer, and 0.1μm p-GaN contact layer.

The surface morphologies of the conventional and the new laser structures are shown in Fig.2. The conventional laser structure has many cracks due to the lattice mismatch between GaN and Al_{0.07}Ga_{0.93}N, while the new structure does not have any crack. From the X-ray diffraction measurement, we have obtained the tensile strain in n-Al_{0.07}Ga_{0.93}N layer of new structure is about 0.06%, which value is almost the half in the case of conventional structure. This lower tensile strain could contribute to few cracks in the wafer irrespective of thicker n-Al_{0.07}Ga_{0.93}N layers.

DEVICE PERFORMANCE

Using this laser wafer, we fabricated the 2-μm-wide ridge waveguide devices with 500-μm-long cavity. The laser facets were formed by cleaving along (1-100) plane of sapphire. High reflection coating was performed on the facets (front:80%, rear: 90%).

Figure 3 shows L-I characteristics of the new type laser. The measurements were pursued at room temperature under pulsed condition with a duty ration of 0.05 %. The threshold current of this device is as low as 65mA, which corresponds to threshold current density of 6.5 kA/cm². This value is equal

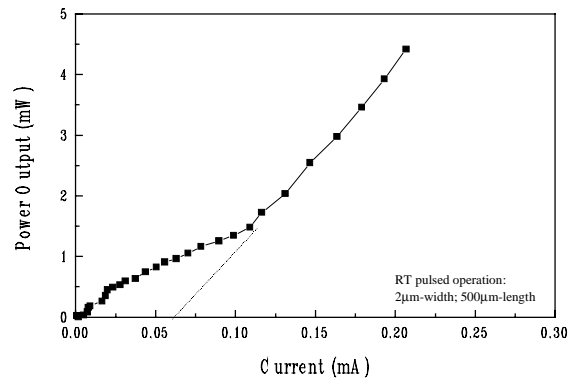


Figure 3: L-I Characteristic under RT Pulsed Condition

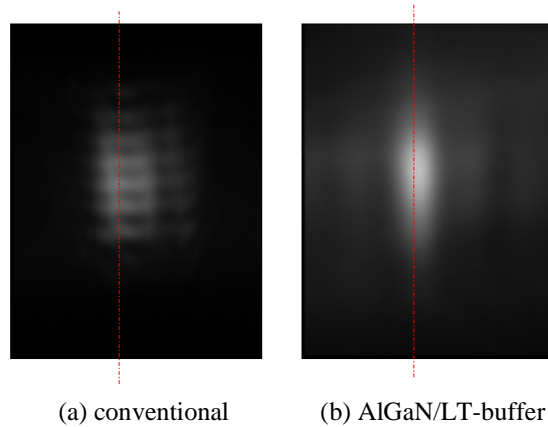


Figure 4: Far field pattern of LDs

to or better than that of conventional one in our experiments, indicating that the material quality of layers on underlying AlGaIn layer / LT-buffer / sapphire is high enough in terms of device performances. The vertical profiles of far field pattern for the conventional and the new type lasers are shown in Fig. 4. The new type laser clearly shows single peak in far field pattern, while the conventional one shows multi peak in far field pattern as other groups already reported.[10],[11] These results indicate that the thick AlGaIn cladding layer can successfully suppress the optical leakage from the waveguide region to the other regions. We also plotted the theoretical profiles of far field patterns calculated. The calculated results are good agreement with the experimental results.

CONCLUSION

In summary, we have realized thick layer of AlGaIn free from cracks, using AlGaIn / LT-buffer layer structure. Applying the structure to laser diode, we have demonstrated a low threshold current operation and a single spot near field and far field patterns in the direction perpendicular to the junction. A $1\mu\text{m}$ $\text{Al}_{0.07}\text{Ga}_{0.93}\text{N}$ cladding layer attributes to suppress the optical leakage and results in a improvement of optical confinement. This structure should be the key technology for practical use of high-performance GaN-based laser diode.

ACKNOWLEDGEMENT

This work was partly supported by the JSPS Research for the Future Program in the Area of Atomic Scale Surface and Interface Dynamics under the project of “Dynamic Process and Control of the Buffer Layer at the Interface in a Highly-Mismatched System” and Ministry of Education, Science, Sports and Culture of Japan (High-Tech Research Center Project and contract nos. 11450131).

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