Nanoparticles and nanotubes induced by femtosecond lasers

S. ELIEZER, ^{1,2} N. ELIAZ, ³ E. GROSSMAN, ¹ D. FISHER, ^{1,4} I. GOUZMAN, ¹ Z. HENIS, ¹ S. PECKER, ¹ Y. HOROVITZ, ¹ M. FRAENKEL, ¹ S. MAMAN, ¹ V. EZERSKY, ⁵ AND D. ELIEZER ⁵

(RECEIVED 6 September 2004: ACCEPTED October 29 2004)

Abstract

In this paper, we suggest the creation of a nanoparticles and nanotubes by using the interaction of a femtosecond laser with a solid target in a vacuum. A simple model is used to predict the optimal target and the laser parameters for the production of efficient nanoparticles. At the Soreq laboratory, experiments are performed with aluminium and carbon targets using a femtosecond laser. The irradiated targets are composed of either a thin layer of aluminium or of carbon, deposited on a transparent heat-insulating glass substrate. The nanoparticle debris is collected on a silicone wafer for X-ray diffraction (XRD), for scanning electron microscopy (SEM), and for atomic force microscopy (AFM). For transmission electron microscopy (TEM), the debris is caught on a copper grid covered on one side with a carbon membrane. Our experiments confirm the creation of crystal nanoparticles for aluminium and nanotubes for carbon experiments.

Keywords: Femtosecond lasers; Nanoparticles; Nanotubes

1. INTRODUCTION

A particle with a dimension larger than about 1 nanometer (the size of one molecule) and smaller than 100 nanometers (order of magnitude for a wavelength of light) is defined as a nanoparticle. These particles are of great interest (Dresselhaus *et al.*, 1996) because of the physics associated with them, namely, the nanoparticles are described by the intermediate regime between quantum physics and classical physics. Furthermore, they possess a large surface to volume ratio. These same particles may also prove important and useful to many industrial and biological applications.

Thus, the synthesis and study of nanoparticles (NP) of various elements and compounds is of great interest both in technological applications and for fundamental research. Traditionally, NPs are produced (Dutta & Hofmann, 2003) by techniques such as arc discharge, vapor deposition, electrochemical deposition, and laser ablation through long pulse duration (about a few ns) in an appropriate gas atmo-

sphere. For example, it has also been suggested (Gamaly *et al.*, 2000) that by using high repetition rate of ultrafast laser pulses, one can obtain carbon films in the presence of an argon gas. In this model, only a low number of atoms are evaporated per single laser pulse.

In our present work, we created nanoparticles (Eliezer et al., 2004) and for the first time nanotubes by a femtosecond laser; solid target interaction. These ultrashort laser pulses allow the completion of the energy deposition into the target well before the target expansion begins, thus effectively decoupling these two stages of the process. The laser energy is initially deposited in the electron subsystem, within a target surface layer of a few tens of nm thick. This stage continues during the laser pulse duration. After the pulse terminates and before the expansion sets in, the electrons in the surface layer undergo cooling by heat diffusion and by energy transfer to the ions. This stage continues for several picoseconds followed by a cooling process that lasts about 1 ns. If during this cooling, a mixture of liquid plus vapor is obtained, then the particles condensate into nanoparticles or nanotubes. The timescales and appropriate dimensions of this scenario can be described by the following:

¹Soreq NRC, Yavne, Israel

²Institute of Nuclear Fusion, Polytechnic University of Madrid, Madrid, Spain

³Tel Aviv University, Tel Aviv, Israel

⁴Weizmann Institute of Science, Rehovot, Israel

⁵Ben-Gurion University, Beer Sheva, Israel

 $Address\ correspondence\ and\ reprint\ requests\ to:\ Shalom\ Eliezer.\ E-mail:\ eliezer@soreq.gov.il$

S. Eliezer et al.

$$\begin{split} \tau_1 &\equiv \text{Energy Deposition} \approx 10^{-13} \left[s \right] \\ &\Rightarrow c_s \tau_1 \approx 5 \cdot 10^{-8} \left[cm \right], \\ \tau_2 &\equiv \text{Thermal Relaxation} \left(T_e = T_i \right) \approx 10^{-12} \left[s \right] \\ &\Rightarrow c_s \tau_2 \approx 5 \cdot 10^{-7} \left[cm \right], \\ \tau_3 &\equiv \text{Nanoparticle production} \approx 10^{-9} \left[s \right] \\ &\Rightarrow c_s \tau_3 \approx 5 \cdot 10^{-4} \left[cm \right], \end{split}$$

where c_s is of the order of speed of sound, taken in the above equation as 5×10^5 cm/s. Note that at room temperature, the speed of sound for aluminum is 5.4×10^5 cm/s, for graphite carbon it is 2.63×10^5 cm/s, while for diamond it is 1.96×10^5 cm/s. It is interesting to see that when a femtosecond laser interacts with a solid target, nanoparticles can be created in a few nanoseconds during which time the material expands into the vacuum at a distance of a few microns. For adiabatic expansion into the vacuum, one has to replace c_s by a velocity of about ten times greater, thus increasing the estimations in Eq. (1) by a factor of ten. Even in this case, the nanoparticles and nanotubes are created during the plasma expansion in less than $100 \ \mu m$.

2. THE LASER PLASMA INTERACTION

In our experiments, the nanoparticles are collected on a silicon crystal at a distance of about 3 cm from the target, therefore allowing free expansion of nanoparticles into the vacuum. These experiments, described schematically in Fig. 1, are carried out with a titanium sapphire laser with 0.8 μm wavelength, 50 fs pulse duration, 0.5 joule of energy, with intensities varying between 3 \times 10 12 W/cm² and 5 \times 10 15 W/cm² . The irradiated targets are composed of a 100 nm thick foil either of aluminum or of carbon, deposited on a transparent heat-insulating glass substrate (e.g., CaF2). The NP debris is collected on a silicone wafer.

Since the target does not change its density during the short laser plasma interaction, one can neglect the hydrodynamic effects and write the following energy conserva-

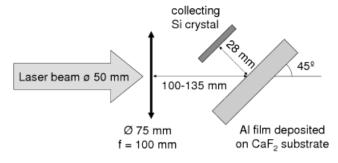


Fig. 1. A schematic view with experimental description of the experiment.

tion in order to calculate the plasma temperature (Fisher *et al.*, 2001; Eliezer, 2002).

$$C_{e}(T_{e}) \frac{\partial T_{e}}{\partial t} = \frac{\partial}{\partial x} \left[\kappa(T_{e}) \frac{\partial T_{e}}{\partial x} \right] - U(T_{e}, T_{i}) + Q(x, t),$$

$$C_{i}(T_{i}) \frac{\partial T_{i}}{\partial t} = U(T_{e}, T_{i}), \tag{2}$$

where T_e and T_i is the electron and ion temperatures, C_e and C_i are the electron and ion heat capacities, κ is the electron heat conductivity, U is the heat transfer rate from electrons to ions, t is time, x is the distance from the target surface inward, and Q is the heat deposition rate due to the laser radiation absorption. The laser absorption changes significantly on the scale of the skin depth, in the order of 10 nm. The laser deposition energy Q is obtained from the solution from Maxwell equations. The detailed expressions for C_e , C_i , and U are given in Fisher $et\ al.\ (2001)$.

3. THE EXPERIMENTAL RESULTS FOR ALUMINUM TARGETS

The experimental setup and conditions are described in Fig. 1. The pressure in the vacuum chamber is 10^{-4} Torr. The laser absorption coefficient changes between 16% to about 40% for laser irradiances of 5×10^{12} W/cm² and 2×10^{14} W/cm² accordingly.

The density and temperature (ρ, T) diagram of aluminum is presented in Fig. 2. Here we are using the accurate equation of state (EOS), similar to the one we developed when analyzing the ultimate strength of materials (Eliezer *et al.*, 2002). The initial conditions of the laser created plasma is given by the solid density ρ_0 and a temperature T_0 determined by the laser plasma interaction. The area between the spinodal and binodal on the right side in Fig. 2 is the super heated liquid domain. If during the adiabatic expan-

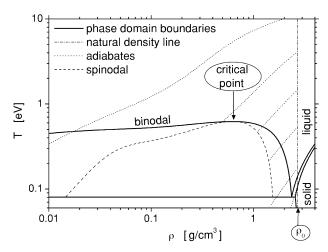


Fig. 2. Phase diagram of Aluminum.

sion from the initial condition ($\rho_0 T_0$), the plasma enters the super heated liquid domain, then the liquid phase is achieved followed by a fast crystallization of the aluminum. Adiabatic paths starting at T_0 in the domain of 0.2 eV to 4 eV are the most promising in the context of nanoparticle productions; since they reach the spinodal lower than the critical density where a large fraction of the material transforms into a liquid state. For initial temperatures T_0 is larger than 10 eV, the plasma expands, cools, recombines into gas, and is eventually deposited on the vacuum vessel walls as individual atoms.

For the appropriate initial conditions, the nanoparticles are created during a period of the order of 1 nanosecond (see Introduction). These nanoparticles expand into the vacuum perpendicular to the aluminum target and are collected on a silicon wafer for X-ray diffraction (XRD), for scanning electron microscopy (SEM), and for atomic force microscopy (AFM). For transmission electron microscopy (TEM), the debris is caught on a copper grid covered on one side with a carbon membrane.

A typical atomic force microscope image (out of many images of different scales) of the nanoparticles deposited onto a silicone substrate is given in Fig. 3. In this particular figure, the scan size in the x-y is 250×250 nm, and 25 nm in the z direction. This image reveals particles ranging from 10 to 50 nm in diameter and from 2 to 10 nm in height.

The size distribution of the nanoparticles is determined with the aid of an image analysis software package based on the AFM data. The cumulative number density f(S) of nanoparticles with an area from ∞ to S (in units of nm²) fits well the following formula:

$$f(S)[particles/\mu m^2] = \alpha_1 \exp\left(-\frac{S}{\mu_1}\right) + \alpha_2 \exp\left(-\frac{S}{\mu_2}\right),$$

$$N[particles/\mu m^2] = f(0) = \alpha_1 + \alpha_2,$$

$$\langle S \rangle = \frac{\alpha_1 \mu_1 + \alpha_2 \mu_2}{\alpha_1 + \alpha_2},\tag{3}$$

where *N* is the total number of particles per unit area and $\langle S \rangle$ is the average area of the nanoparticles.

Our experiments show the following data: N=4, 300, 1000, and 30 [particles/ μ m²] for $I_L=3\times10^{12}$, 3.7×10^{13} , 5.3×10^{13} , and 5×10^{14} W/cm², respectively. For the same laser irradiances we get appropriately $\langle S\rangle=1.2\times10^4$, 4.2×10^2 , 1.2×10^2 , and 5×10^3 nm². These experimental values are in agreement with the qualitative model described above.

The TEM data characterizes the structure and the composition of the nanoparticles at a subnanometer resolution. Line arrays of atoms are measured indicating the crystalline nature of the aluminum nanoparticles. The distance between the atomic layers of the crystal is found to be 2.354 \pm 0.118 Å, which is in good agreement with the strongest reflection from the cubic Al phase (2.338 Å for planes {111}). It is important to point out that these crystals contain less than 10 atoms in each direction.

4. THE CARBON TARGETS

For carbon the EOS is more complicated and we do not have all of the necessary data and knowledge to make a rigorous analysis like the one for aluminum in the previous section.

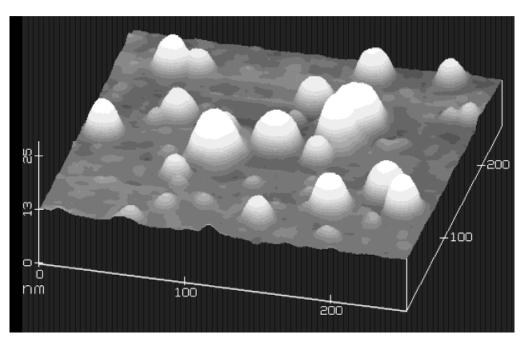


Fig. 3. An atomic force microscope (AFM) image for Aluminum nanoparticles.

18 S. Eliezer et al.

Therefore, a simplified analysis is done here for the carbon case.

The energy equation used in this simplified analysis equates the absorbed laser energy to the integrated heat capacity,

$$(1-R)(F-F_m)\alpha = nc_P(T-T_b). \tag{4}$$

The nanotubes are obtained with a laser fluency of F = 7.5 J/cm², implying a laser irradiance of $I = 1.5 \times 10^{14}$ W/cm². The fluency of the femtosecond laser that melts (actually sublimates) the graphite is F_m J/cm² = 0.13 (Reitze *et al.*, 1992). From the reflection experiments, one gets for our case R = 0.2, namely 80% of the laser energy is absorbed. The $\alpha^{-1} = 10^{-5}$ cm is the carbon layer thickness, assumed to have an initial constant temperature T, $T_b = 4500$ K is the boiling temperature, $c_P = 3$ k_B (k_B is the Boltzmann constant) is the Dulong-Petit value of the heat capacity and $n = 1.12 \times 10^{23}$ cm⁻³ is the initial density of the graphite plasma (for $\rho_0 = 2.25$ g/cm³). Eq. (4) yields an initial temperature of T = 111,000 K

Our laser focal spot is large compared to the plasma expansion before it solidifies. Therefore, it is conceivable to assume a one dimensional (1D) adiabatic expansion described by the following equations:

$$\frac{T}{T_b} = \left(\frac{\rho}{\rho_0}\right)^{\gamma - 1} = \left(\frac{x_0}{x_0 + ut}\right)^{\gamma - 1}$$

$$u = \frac{2}{\gamma - 1} \left[\frac{Zk_B T}{M}\right] \approx 2 \cdot 10^6 [T(eV)]^{1/2} \left[\frac{cm}{s}\right], \tag{5}$$

where $x_0 = \alpha^{-1}$ is the graphite foil thickness, $\gamma = 1.6$ is the adiabatic constant, u is the plasma expansion velocity, Z = 4 is the appropriate ionization of carbon and M is the carbon mass. For the temperature derived in Eq. (4), one gets $u = 6 \times 10^6$ cm/s. This value yields from Eq. (5) a time t = 0.35 ns, namely the plasma has cooled to the boiling temperature T_b in a time of 0.35 ns while expanding only 60 μ m. We don't know the time required for the nanotube productions; however, it is estimated to be in the order of one to a few nanoseconds.

Figures 4 and 5 were taken with a TEM. Fig. 4 shows high resolution onion-like graphitic particles (Dresselhaus *et al.*, 1996). The concentric spherical shells, the so called "onions," constitute a form of carbon consisting of spherical concentric layers of carbon shells. In Fig. 5, about 10 concentric tubes are seen when the internal tube has a diameter of about 5 nm.

5. CONCLUSIONS

The main methods used to synthesize nanotubes are by carbon arc discharge, chemical vapor deposition, and ion bombardment. Nanoparticles have also been achieved with these techniques and also by laser ablative deposition in an appropriate gas atmosphere. In this paper, it has been sug-

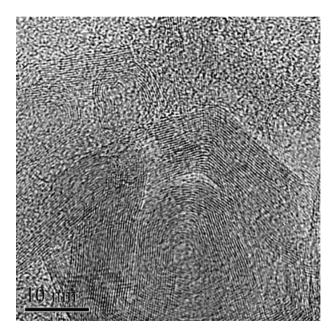


Fig. 4. Onion like carbon structures as measured by TEM.

gested to use the femtosecond lasers to create nanocrystals and nanotubes in a vacuum medium. One of the advantages of this scheme is that the interaction of the external field (the laser in our case) with the material under consideration during the nanoparticle or nanotube formation is non existent. The femtosecond laser is used only in order to achieve appropriate initial conditions. Therefore, in these experiments it might be easier to understand the mechanism of the production of nanotubes and nanocrystals. Furthermore, with the laser, there are no limits on the density and temper-

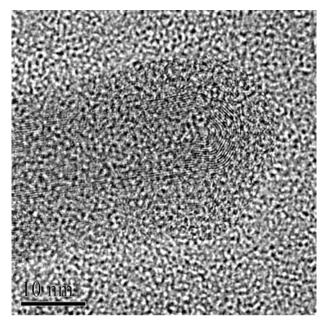


Fig. 5. Nanotubes of carbon as measured by TEM.

ature of the initial conditions. Very high temperatures and solid densities can be easily achieved. More work and experiments are required in order to comprehend the new ideas suggested in this work.

REFERENCES

- Dresselhaus, M.S., Dresselhaus, G. & Eklund, P.C. (1996). Science of Fullerenes and Carbon Nanotubes. San Diego, CA: Academic Press.
- DUTTA, J. & HOFMANN, H. (2003). *Nanomaterials*. http://www.nano.ait.ac.ch.
- ELIEZER, S. (2002). *The Interaction of High Power Lasers with Plasmas*. Bristol, UK: Institute of Physics.
- ELIEZER, S., MOSHE, E. & ELIEZER, D. (2002). Laser-induced

- tension to measure the ultimate strength of metals related to the equation of state. *Laser Part. Beams* **20**, 87.
- ELIEZER, S., ELIAZ, N., GROSSMAN, E., FISHER, D., GOUZMAN, I., HENIS, Z., PECKER, S., HOROVITZ, Y., FRAENKEL, M., MAMAN, S. & LEREAH, Y. (2004). Synthesis of nanoparticles with femtosecond laser pulses. *Phys. Rev. B* **69**, 144119.
- FISHER, D., FRAENKEL, M., HENIS, Z., MOSHE, E. & ELIEZER, S. (2001). Interband and intraband (Drude) contributions to femtosecond laser absorption in aluminum. *Phys. Rev. E* **65**, 016409.
- Gamaly, E.G., Rode, A.V. & Luther-Davies, B. (2000). Formation of diamond-like carbon films and carbon foam by ultrafast laser ablation. *Laser Part. Beams* 18, 245.
- REITZE, D.H., AHN, H. & DOWNER, M.C. (1992). Optical properties of liquid carbon measured by femtosecond spectroscopy. *Phys. Rev. B* **45**, 2677.